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# Unraveling active sites regulation and temperature-dependent thermodynamic mechanism in photothermocatalytic $CO_2$ conversion with $H_2O$

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In the photothermal synergistic catalytic conversion of CO<sub>2</sub> and H<sub>2</sub>O, the catalyst harnesses solar energy to accumulate heat, thereby elevating the reaction system's temperature. The influence of this temperature effect on surface chemical reactions remains an underexplored area. Here the impact of temperature on the surface-level thermodynamic reactions and conversion of CO<sub>2</sub> with H<sub>2</sub>O on oxide semiconductors at the atomic scale was investigated using first-principle calculations. 13 different metal oxides and 5 transition metal clusters were used to introduce surface functional sites on the  $TiO_2$ supporting catalyst. The potential metal oxide cocatalysts that could be most beneficial to the following conversion of CO<sub>2</sub> by H<sub>2</sub>O were initially screened by calculating the degrees of promotion of CO<sub>2</sub> adsorption and activation of surface H to provide protons. The proton donation and hydrogen evolution difficulty from H<sub>2</sub>O were further analyzed, identifying transition metal cocatalysts that promote direct CO<sub>2</sub> hydrogenation. Upon introducing bifunctional sites to facilitate adsorption and reduction, the production of  $CH_3OH$  and  $CH_4$  could be further enhanced through the facilitation of the proton donation process of H<sub>2</sub>O. The results of Gibbs free-energy calculations revealed that increasing temperature enhances the reaction thermodynamics for each C1 product formation at different surface sites to varying degrees. These findings offer valuable theoretical insights for designing and regulating active sites on oxide semiconductor surfaces for efficient photothermal catalytic  $CO_2$  reduction by  $H_2O$ .

Increasing apprehensions concerning global energy demand and greenhouse impacts have fueled the advancement of clean and renewable energy systems. The utilization of solar energy for instigating chemical reactions to convert  $CO_2$  into solar fuels is considered a potential solution, which can simultaneously mitigate the greenhouse effect and ameliorate the energy crisis<sup>1,2</sup>. Photothermal synergistic catalytic conversion of  $CO_2$  with  $H_2O$ , when viewed from a catalytic pathway perspective, amalgamates the benefits of photogenerated carrier-driven photocatalysis and temperature-driven traditional thermal catalysis, and enables the coupling of non-thermal and thermal effects of solar radiation<sup>3</sup>. In terms of the feedstock, the abundance and environmental friendliness of hydrogen as a source make this approach exceptionally promising, and the development in this field is rapidly advancing<sup>4,5</sup>. In the process of advancing studies on  $CO_2$  photothermal conversion, it is imperative to understand the separate roles of light and heat and, subsequently, grasp the synergistic mechanism. Directly elucidating photothermal synergy and systematically assessing the combined effects of light and heat excitation on catalytic conversion has always been a challenging and complex issue<sup>6</sup>.

Several recent reviews have presented varying definitions for the terminology concerning the impacts of light and heat<sup>7-12</sup>, signifying a growing trend in research that seeks to elucidate these effects from a fundamental mechanistic standpoint. The photothermal process is jointly contributed by photochemistry and thermochemical mechanistic pathways. Accordingly, Mateo et al.<sup>7</sup> categorized it as "hot-carrier" and "thermal" effects, while Song et al.<sup>8</sup> referred to it as "light" and "thermal" effects. The more general term, "light" and "heat", are commonly encountered in current research literature<sup>9,10</sup>. Ozin introduced the concepts of "photochemical" and "photophysical" effects<sup>6</sup>, and highlighted the ongoing challenge of determining the relative contributions of these two effects to reaction rate and selectivity. Becker et al.<sup>11</sup> adopted a simplified model for evaluating the

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"photochemical" and "thermochemical" effects in the context of exothermic reactions involving solid catalysts exposed to irradiation. Fang et al.'s review focused on the synergistic effects of "thermal and photo energies" in catalysis<sup>12</sup>. Chen et al.<sup>13</sup> have unveiled the synergistic mechanism involving the coupling of "photo" and "photothermal" effects with specific catalysts like Ru@Ni<sub>2</sub>V<sub>2</sub>O<sub>7</sub>. It should be noted that the temperature effects mentioned here collectively refer to the increase in system temperature after the accumulation of thermal energy caused by thermal effects, which may be attributed to the non-radiative relaxation of semiconductors, localized heating induced by plasma, and thermal vibration of molecules<sup>14</sup>. This is relative to the non-thermal effects involving the generation and utilization of high-energy carriers<sup>15</sup>. The elevation in reaction temperature will significantly influence the entire reaction system, encompassing reaction thermodynamics and kinetics, product selectivity and yield, thermal active sites, and catalyst structure<sup>9,12</sup>. Some recent studies have identified the temperature effects including enhancements in diffusion rates, adsorptiondesorption processes, charge carrier migration, and chemical reactions<sup>16-18</sup>. Temperature also facilitates the surmounting of activation energy barriers<sup>19</sup>, activation of thermally active sites<sup>20</sup>, and adjustments to the redox potential of half-reactions<sup>21,22</sup>, along with alterations in catalyst band gaps<sup>23</sup>. Nevertheless, there remains a lack of atomic-level understanding of the intricate surface chemistry and the thermodynamic reaction path mechanism affected by temperature in the CO<sub>2</sub> catalytic reduction process<sup>24,25</sup>.

While trying to break through research on the mechanisms underlying both non-thermal and thermal effects, scholars turn their eyes toward the target reaction of CO<sub>2</sub> conversion with H<sub>2</sub>O, a pivotal and fundamental reaction in the field of solar energy conversion. It is well-known that the chemical properties of CO2 and H2O are notably stable, necessitating materials characterized by favorable non-thermal effects to surmount the stringent reaction conditions required for the activation of reactant molecules. Numerous conventional metal oxide semiconductors, including TiO<sub>2</sub>, have been utilized for photothermal CO<sub>2</sub> reduction in the presence of pure H<sub>2</sub>O. However, the majority of the reported systems thus far have demonstrated relatively low CO2 conversion rates, typically in the magnitude of µmol·g<sup>-1</sup>·h<sup>-1</sup><sup>26,27</sup>. Based on the basic principles and processes of photothermal catalytic CO<sub>2</sub> reduction by H<sub>2</sub>O, ongoing research efforts on improving CO<sub>2</sub> conversion performance mainly focus on strengthening the following three key steps: 1. Photon absorption; 2. Photogenerated carrier separation; 3. Surface catalytic reactions involving CO2 and H2O (including adsorption of reactants, and desorption and reduction of intermediates, etc.)<sup>28</sup>. For the third step, multiple active sites need to be designed on the catalyst surface to simultaneously achieve CO2 reduction and H2O dissociation, which still remains tremendously challenging<sup>29</sup>. In the photothermal catalytic reaction system where CO2 and H2O coexist, either H2O or H<sub>2</sub> might become proton sources. There are three situations in which CO<sub>2</sub> reduction occurs. One is that H2O first undergoes photooxidation (assisted by photogenerated holes) to produce protons, the second is that H<sub>2</sub>O directly provides protons, and the third is that photogenerated holes also assist H<sub>2</sub> dissociation producing protons. Then protons reduce CO<sub>2</sub> with photogenerated electrons. While the process of H<sub>2</sub>O splitting for hydrogen evolution can supply H<sup>+</sup> to reduce CO<sub>2</sub>, it simultaneously competes with conduction band electrons for CO2 reduction, diminishing the effective reduction of CO<sub>2</sub> by photoexcited electrons<sup>30</sup>. Additionally, H<sub>2</sub>O tends to have a higher propensity for surface adsorption on photocatalysts due to its high polarity. In contrast, initial adsorption and activation of CO2 usually require a huge potential barrier. These result in low CO2 reduction efficiency and poor hydrocarbon product selectivity<sup>31,32</sup>. Surface defect engineering<sup>31</sup> and co-catalyst engineering<sup>34</sup> have important applications in photothermal catalytic CO<sub>2</sub> reduction reactions<sup>35</sup>. In addition to bringing about alterations in the band gap and electronic structure, thereby changing the radiation absorption and conversion performance of the material, the introduction of oxygen vacancies can also function as reaction active sites for the adsorption and activation of surface species during the catalytic process<sup>36,37</sup>. Similarly, the incorporation of metal oxides and the loading of nanometal particles can further create tailored functional sites on the material surface, enhancing the selective adsorption and activation of reactants and facilitating the subsequent reduction of key intermediates<sup>38,39</sup>. Both approaches have been proven to be powerful means to improve the performance of CO<sub>2</sub> reduction by pure H<sub>2</sub>O on the surface of oxide semiconductors<sup>40</sup>. Alkaline-earth metal oxides including MgO, CaO, SrO, BaO<sup>41,42</sup>, as well as rare-earth metal oxides such as La<sub>2</sub>O<sub>3</sub>, CeO<sub>2</sub>, Pr<sub>6</sub>O<sub>11</sub><sup>43–45</sup> have been widely used as adsorption cocatalysts to introduce basic active sites on the surface of support oxides. Other metal oxides like Al<sub>2</sub>O<sub>3</sub><sup>46</sup>, Ga<sub>2</sub>O<sub>3</sub><sup>47</sup>, ZrO<sub>2</sub><sup>48</sup>, Ta<sub>2</sub>O<sub>5</sub><sup>49</sup>, HfO<sub>2</sub><sup>50</sup>, and WO<sub>3</sub><sup>51</sup> have also been experimentally explored for their potential in photocatalytic reduction of CO<sub>2</sub> and H<sub>2</sub>O, which show promise as promoters to enhance catalytic activity by enriching surface-active sites or modifying the electronic structure. Additionally, Mn, Fe, Co, Ni, and Cu are relatively lowcost representative transition metals capable of activating CO<sub>2</sub> and H<sub>2</sub><sup>52-54</sup>.

This work aims to reveal the effect of temperature on the surface reaction path mechanism coupling CO2 reduction and H2O decomposition involving multiple surface intermediates and hydrodeoxygenation steps. Anatase TiO<sub>2</sub> (aTiO<sub>2</sub>) served as the supporting catalyst, and successive modifications were made on its surface using 13 different metal oxides (La2O3, Al2O3, Ga2O3, CaO, MgO, SrO, CeO2, HfO2, ZrO2, BaO, Ta<sub>2</sub>O<sub>5</sub>, Pr<sub>6</sub>O<sub>11</sub>, WO<sub>3</sub>) and 5 transition metal (Mn, Fe, Co, Ni, and Cu) clusters. Subsequently, the adsorption and conversion processes of CO2 and H2O at the distinct surface adsorption and reduction sites correspondingly introduced through these modifications were studied. It was found that pure Ti-O-M sites can promote CO2 adsorption, but do not necessarily facilitate subsequent reduction reactions. The metal oxide cocatalysts that may be most beneficial to the following conversion of CO<sub>2</sub> were initially screened as La<sub>2</sub>O<sub>3</sub>, Al<sub>2</sub>O<sub>3</sub>, and Ga<sub>2</sub>O<sub>3</sub>, which can not only promote CO<sub>2</sub> adsorption but also activate surface H to provide protons. Subsequent loading of Co4 and Ni4 transition metal clusters further promoted the proton-donating process of H<sub>2</sub>O splitting while inhibiting hydrogen evolution, and are expected to improve the CH<sub>3</sub>OH and CH<sub>4</sub> selectivity by further promoting the direct hydrogenation of CO<sub>2</sub>. In addition, the results of Gibbs free-energy calculations indicate that increasing temperature enhances the thermodynamics of C1 product formation on the surface at different sites to varying degrees.

#### **Results and discussion**

The CO<sub>2</sub> photoreduction with H<sub>2</sub>O on the surface of metal oxide semiconductor involves sophisticated multi-reaction paths, accompanied by multiple proton-coupled electrons transfer<sup>55</sup>, and protonation, deoxygenation, and hydroxylation steps. It results in a variety of intermediate species and a complex reaction process. The elaborated thermodynamic reaction network of CO<sub>2</sub> conversion with H<sub>2</sub>O on aTiO<sub>2</sub>(101)-based surface has been reported in our previous work<sup>56</sup>. To observe its temperaturedependent characteristics, here the reaction paths of CO<sub>2</sub> with H<sub>2</sub>O to C<sub>1</sub> products (mainly CO, CH<sub>3</sub>OH and CH<sub>4</sub>) on pure aTiO<sub>2</sub>(101) surface at 100~400 °C were calculated, and the energy barriers comparison of all intermediate steps were shown in Supplementary Fig. 3a. During the process of photothermal catalytic CO<sub>2</sub> conversion with H<sub>2</sub>O, both H<sub>2</sub>O and H<sub>2</sub> might provide proton sources and reduce CO2 together with photogenerated electrons. Here we focused on the forward protonation steps of  $CO_2$  conversion (as shown in Fig. 1). It was found that, on pure  $aTiO_2(101)$ surface, the energy barriers of certain intermediate steps exhibited an increase as temperature ascends, while others displayed a decrease. However, the fundamental trajectory of the reaction remained unaffected by changes in temperature. The rate-determining step (RDS) of producing CO, CH<sub>3</sub>OH, and CH<sub>4</sub> was always the CO<sub>2</sub> adsorption step but with its energy barrier increased from 0.41 eV to 1.04 eV (as shown in Supplementary Fig. 3b), which might be attributed to the fact that the adsorption is exothermic commonly while the increasing temperature would inhibit the reactants adsorption. Therefore, it means that reactant adsorption under hightemperature conditions plays a critical role in the whole reaction. Optimizing surface adsorption sites will be considered first. Furthermore, the subsequent conversion of reactants on the catalyst also exhibited a strong correlation with the presence of active sites on the surface. Basically,

introducing adsorption sites and reduction sites on the support semiconductors has been regarded as an effective approach to enhance the performance of photothermal catalytic  $\rm CO_2$  reduction<sup>57</sup>. In this context, the discussion revolves around elucidating the mechanism that governed the influence of temperature on the thermodynamic reaction process path (including the adsorption and conversion of reactants and intermediates) and product selectivity during the photothermal catalysis of  $\rm CO_2$  conversion with H<sub>2</sub>O on semiconductor surfaces featuring functional active sites.



Fig. 1 | The protonation steps and reaction network of CO<sub>2</sub> conversion with H<sub>2</sub>O to C<sub>1</sub> products. During the photothermal catalytic CO<sub>2</sub> conversion with H<sub>2</sub>O, light irradiates H<sub>2</sub>O molecules to split and provide protons and electrons to reduce CO<sub>2</sub>. It takes two protons and two electrons to generate CO, six protons and two electrons to generate CH<sub>3</sub>OH, and eight protons and two electrons to generate CH<sub>4</sub>.



Fig. 2 | Comparison of the difficulty of adsorbing and activating  $CO_2$ ,  $H_2O$  molecules, H atom, and forming  $V_O$  on TiO<sub>2</sub> surfaces before and after  $M_xO_y$  decorating. a Energy barriers ( $E_{ads}$ ) for activating  $CO_2$ ,  $H_2O$  molecules, and H atoms

#### Adsorption and activation of CO<sub>2</sub>, H<sub>2</sub>O, and H atom

The pure  $aTiO_2(101)$  surface owns relatively simple reaction sites for reactant adsorption and activation. For the purpose of designing and providing selective sites for the adsorption of CO2, H2O, and H, 13 types of M<sub>x</sub>O<sub>y</sub> (La<sub>2</sub>O<sub>3</sub>, Al<sub>2</sub>O<sub>3</sub>, Ga<sub>2</sub>O<sub>3</sub>, CaO, MgO, SrO, CeO<sub>2</sub>, HfO<sub>2</sub>, ZrO<sub>2</sub>, BaO, Ta<sub>2</sub>O<sub>5</sub>, Pr<sub>6</sub>O<sub>11</sub>, WO<sub>3</sub>) were decorated on the pure aTiO<sub>2</sub>(101) surface. As demonstrated in Fig. 2a, the energy barriers for adsorption and activation of CO2 and H2O molecule, H atom on surface Ti-O (VO)-M sites were calculated. It can be concluded that both photoinduced V<sub>O</sub> and doped M<sub>x</sub>O<sub>y</sub> promoted CO<sub>2</sub> adsorption to varying extents. Compared with this, H<sub>2</sub>O adsorption was slightly affected, which emphasized the favorable effects of surface oxygen defects and basic sites on promoting the initial activation and selective adsorption of CO<sub>2</sub>. Regarding the adsorption of H atoms, La<sub>2</sub>O<sub>3</sub>, Al<sub>2</sub>O<sub>3</sub>, and Ga<sub>2</sub>O<sub>3</sub> doping exerted the most significant influence among all modified metal oxides. On the Ti-O-M (M=La, Al, Ga) site, the adsorption energy of H attained a notably diminished value. The charge density difference analysis was conducted for H adsorption on M<sub>x</sub>O<sub>y</sub>-decorated surfaces. As depicted in Supplementary Fig. 4, the electron gains and losses between the H atom and the surface were notably pronounced on surfaces doped with La2O3, Al2O3, and Ga2O3, with an increased charge transfer observed. A higher level of activation of H on the surface could potentially enhance the favorability of CO<sub>2</sub> hydrogenation. It should be noted that the difficulty in creating oxygen vacancies on the surface post-doping with distinct oxides exhibited differing degrees, and the calculated oxygen vacancy formation energy was presented in Fig. 2b for reference. Broadly speaking, when the valence state of the doped oxide's metal cations M<sup>2y/x+</sup> increased, detaching surface oxygen became more difficult, which might be attributed to the more pronounced charge compensation effect stemming from low-valent metal cations<sup>58</sup>. On surfaces doped with CaO, MgO, SrO, and BaO, the formation energy of V<sub>O</sub> experienced a significant reduction, leading to the ease of forming Ti-Vo-M sites. This made reactant molecules and atoms exhibit a propensity for adsorption and activation directly on the defective surface. Comparatively, the presence of photoinduced Vo and the introduction of La2O3, Al2O3, and Ga2O3 on the aTiO2 surface demonstrated a relatively potent promotional effect on CO2 activation. Additionally, during photothermal catalysis reactions, light irradiation triggered the generation of oxygen vacancies on the material's surface. Under these circumstances, the adsorption energy of CO2 on the surface of La2O3, Al2O3, and Ga2O3-doped TiO2 was indicated in Supplementary Table 1. Notably, the adsorption energy of  $CO_2$  evolved from -1.20 eV, -0.77 eV, and -0.76 eV (in the absence of surface defects) to -0.61 eV, -0.52 eV, and



to CO2  $^*_2,$  H2O  $^*_2,$  and H  $^*_2,$  respectively, on different surface sites. **b** Formation energy of V0 on corresponding surfaces.



Fig. 3 | Reaction free-energy profiles of thermodynamic reaction pathways for multi-electron conversion of CO<sub>2</sub> to C<sub>1</sub> (CO, CH<sub>3</sub>OH, and CH<sub>4</sub>) products at varying temperatures (100~400 °C). Different reaction pathways and key

intermediate steps are experienced on the surfaces of **a** aTiO<sub>2</sub> with V<sub>O</sub>, **b** Al<sub>2</sub>O<sub>3</sub>-doped aTiO<sub>2</sub> with V<sub>O</sub>, **c** La<sub>2</sub>O<sub>3</sub>-doped aTiO<sub>2</sub> with V<sub>O</sub>, and **d** Ga<sub>2</sub>O<sub>3</sub>-doped aTiO<sub>2</sub> with V<sub>O</sub> that strengthen the adsorption of reactants.

-0.52 eV respectively, signifying an increase yet maintaining a promotional effect. The enhancement of the basic sites introduced via doping for subsequent CO<sub>2</sub> conversion reactions relied on the availability of protons and electrons. The collaborative presence of synergistic surface oxygen defects and basic sites worked in tandem to amplify the reaction's thermodynamics, facilitating the multi-electron conversion of CO<sub>2</sub> to yield C1 products<sup>56</sup>. Based on the efficacy of La<sub>2</sub>O<sub>3</sub>, Al<sub>2</sub>O<sub>3</sub>, and Ga<sub>2</sub>O<sub>3</sub> in augmenting the adsorption and activation of CO<sub>2</sub> molecules and H atoms, comprehensive calculations of the entire thermodynamic reaction paths for photothermal catalytic CO<sub>2</sub> conversion to C<sub>1</sub> products (CO, CH<sub>3</sub>OH, and CH<sub>4</sub>) at varying temperatures were performed on the respective defective surfaces doped with these compounds.

# Temperature influence on the thermodynamic path of surface $CO_2$ conversion via enhanced adsorption sites

Thermodynamically, adsorption is exothermic, with a typically negative temperature impact. In other words, as the temperature increases, the strength of adsorption decreases. Yet, when examining surface CO2 conversion reactions that encompass various intermediate steps, the interplay between temperature and these reactions becomes notably intricate compared to the simpler phenomenon of adsorption. On the surface-active sites that strengthen the adsorption of reactants, the reaction pathways of CO<sub>2</sub> conversion to CO, CH<sub>3</sub>OH, and CH<sub>4</sub> at varying temperatures were illustrated in Figs. 3 and 4. As shown in Figs. 3a and 4a, on the defective aTiO<sub>2</sub> surface, the RDS of each product was different from that on the pure aTiO<sub>2</sub> surface. In the process of CO generation through  $CO_2 \rightarrow CO_2 \rightarrow COOH \rightarrow CO \rightarrow CO'$ ,  $CO_2$  adsorption was no longer the RDS on the defective surface. Instead, the RDS could involve either CO desorption or protonation of <sup>\*</sup>CO<sub>2</sub> to form <sup>\*</sup>COOH. With a temperature increased from 100 °C to 150 °C, CO desorption became the dominant RDS for CO production, resulting in a reduction of the associated energy barrier (from 0.49 eV to 0.31 eV). However, upon further elevation to 200 °C, the RDS transitioned to '\*CO<sub>2</sub> protonation to \*COOH', accompanied by an energy barrier increase (from 0.27 eV to 0.38 eV) as temperature rose. Similarly, the RDS for CH<sub>3</sub>OH production underwent alterations (from '\*CH<sub>3</sub>OH→CH<sub>3</sub>OH' to '\*CHO→\*HCOH'), and the energy barrier first decreased and then increased. This suggested the existence of an optimal thermodynamic temperature range for CO and CH<sub>3</sub>OH generation, specifically between 150 °C and 200 °C. Meanwhile, the energy barrier associated with the RDS (\*CHO→\*HCOH) for CH<sub>4</sub> production showed an upward trend with increasing temperature, eventually being consistent with the energy barrier observed for CH<sub>3</sub>OH production.

Across defective surfaces featuring various introduced basic sites, the reaction paths of C1 products exhibited discrepancies. Different surfaces tended to favor distinct optimal reaction paths, and engaged diverse reaction intermediates, as presented in Supplementary Fig. 5. When the temperature increased, notable on the Al<sub>2</sub>O<sub>3</sub> doped surface as shown in Figs. 3b and 4b, the RDS for CO production experienced a sequential transformation: transitioning from '\*COOH $\rightarrow$ \*CO' to '\*CO<sub>2</sub> $\rightarrow$ \*COOH', and finally culminating in CO2 adsorption. This sequential evolution was accompanied by an initial reduction followed by an eventual increase in the associated energy barrier. Notably, a thermodynamically favorable temperature range for CO production emerged, peaking around 150 °C. Likewise, for CH<sub>3</sub>OH production, the RDS shifted from <sup>°</sup>CH<sub>3</sub>OH→CH<sub>3</sub>OH' to <sup>°</sup>CH<sub>3</sub>O→<sup>\*</sup>CH<sub>3</sub>OH' with increasing temperature, leading to an initial energy barrier decrease followed by an increase. The most thermodynamically favorable temperature for CH<sub>3</sub>OH production was approximately 250 °C. Conversely, the energy barrier for the RDS of CH<sub>4</sub> production ( $^{*}CH_{3}O \rightarrow ^{*}CH_{3}OH^{2}$ ) increased consistently with temperature. On the La2O3 doped surface as shown in Figs. 3c and 4c, the RDS of CO production was always "CO2



**Fig. 4** | **Comparison of temperature-dependent alterations in the RDS energy barriers for CO, CH<sub>3</sub>OH, and CH<sub>4</sub> production.** The Gibbs free-energy change of RDS under the temperature range of 100~400 °C on the surface of **a** aTiO<sub>2</sub> with V<sub>O</sub>,

**b** Al<sub>2</sub>O<sub>3</sub>-doped aTiO<sub>2</sub> with V<sub>O</sub>, **c** La<sub>2</sub>O<sub>3</sub>-doped aTiO<sub>2</sub> with V<sub>O</sub>, and **d** Ga<sub>2</sub>O<sub>3</sub>-doped aTiO<sub>2</sub> with V<sub>O</sub>, respectively.

protonation to "COOH' as the temperature rose, but the energy barrier continued to increase (from 0.98 eV at 100 °C to 1.31 eV at 400 °C), which was not conducive to the formation of CO. The RDS for CH<sub>3</sub>OH and CH<sub>4</sub> production decreased first and then increased. In the temperature range from 100 °C to 200 °C, the pivotal step was the conversion of <sup>\*</sup>H<sub>2</sub>COOH to \*HCHO, and the energy barrier declined with rising temperature, reaching its minimum value of 0.76 eV at 200 °C. Subsequently, as the temperature continued to rise, the RDS transitioned to ' $^{*}HCOO \rightarrow ^{*}HCOOH$ ', suggesting that the most thermodynamically favorable temperature for CH<sub>3</sub>OH and CH<sub>4</sub> production might be approximately 200 °C. In addition, the thermodynamic tendency of the C1 products evolved with increasing temperature, distinct from both undoped and Al<sub>2</sub>O<sub>3</sub>-doped surfaces. As shown in Fig. 4c, at lower temperatures (approximately lower than 50 °C), the energy barrier for CO production was the lowest, suggesting a thermodynamic preference for CO generation at this time. While as the temperature rose, the RDS energy barrier for CO production increased until it surpassed that for CH<sub>3</sub>OH and CH<sub>4</sub> production, and remained high thereafter, leading to increased selectivity for CH<sub>3</sub>OH and CH<sub>4</sub>. With Ga<sub>2</sub>O<sub>3</sub> doping, as shown in Figs. 3d and 4d, an increase in temperature results in a rising energy barrier for the RDS ( $^{*}CH_{3}O \rightarrow ^{*}CH_{3}OH$ ) of CH<sub>4</sub> production. The RDS for CH<sub>3</sub>OH production transitioned from CH<sub>3</sub>OH desorption to <sup>\*</sup>CH<sub>3</sub>O conversion to <sup>c</sup>CH<sub>3</sub>OH', with its energy barrier displaying an initial decrease followed by

an increase. A similar trend was observed for the RDS governing CO production. This transition involved a shift from '\*COOH $\rightarrow$ \*CO' to CO<sub>2</sub> adsorption as the temperature escalated. Furthermore, at temperatures below about 175 °C, the RDS energy barrier for CH<sub>4</sub> production was lower than that for CO and CH<sub>3</sub>OH. However, as the temperature exceeded 175 °C, the energy barrier for CO production decreased below that for CH<sub>3</sub>OH and CH<sub>4</sub>, signifying that this increasing temperature enhanced the thermodynamic inclination toward CO production.

It can be seen from the above that surface doping with basic metal oxides notably influenced the reaction path for  $CO_2$  reduction to  $C_1$  products by altering the reaction intermediates and key steps. The energy barriers of the rate-determining steps for the formation of C1 products corresponding to the different adsorption-assisting active sites introduced changed and were affected by temperature. The higher temperature accentuated the thermodynamic favorability of multi-electron conversion of  $CO_2$  into  $C_1$  products. The temperature variation resulted in shifts in the relative ease of producing CO, CH<sub>3</sub>OH, and CH<sub>4</sub>, as shown in Supplementary Fig. 6. To elaborate, at lower temperatures, the RDS energy barrier for CO production was the lowest on Al-V<sub>O</sub>-Ti site, while the RDS energy barriers for CH<sub>3</sub>OH and CH<sub>4</sub> production were the lowest on pure Ti-O-Ti site, which are lower than those of all defective and doped surfaces. Before 250 °C, it was difficult to produce C1 on La-V<sub>O</sub>-Ti, with the energy barrier

Fig. 5 | Analysis of energy barriers for key steps ( ${}^{*}H_2O \rightarrow {}^{*}OH + {}^{*}H, H \rightarrow {}^{*}H, {}^{*}OH + {}^{*}H \rightarrow {}^{*}O + H_2$ , and  ${}^{*}H + {}^{*}H \rightarrow {}^{*}H_2$ ) in the proton-donating process of H<sub>2</sub>O splitting on different surface-active sites. The upper dotted line of the figure is aligned with the energy barrier of the hydrogen evolution step ( ${}^{*}OH$  $+ {}^{*}H \rightarrow {}^{*}O + H_2$ ) at the pure Ti-O-Ti site, and the dotted line below is aligned with the energy barrier of the water-splitting adsorption step ( ${}^{*}H_2O \rightarrow {}^{*}OH$  $+ {}^{*}H$ ) at the pure Ti-O-Ti site. The surface site where the pink rectangular area is located represents which has the potential to promote water splitting to generate  ${}^{*}H$  and activate H while inhibiting the formation of H<sub>2</sub>.



for CO exceeding that of CH<sub>3</sub>OH and CH<sub>4</sub> and higher than that on Ti-O-Ti site. However, as the temperature rose, Ti-V<sub>O</sub>-Ti site exhibited the lowest RDS energy barrier for CO production, while Ga-Vo-Ti site exhibited the lowest RDS energy barrier for CH<sub>3</sub>OH and CH<sub>4</sub> production, maintaining this status at higher temperatures. The RDS energy barrier for CO production on La-V<sub>O</sub>-Ti remained high, but CH<sub>3</sub>OH and CH<sub>4</sub> production was enhanced. Although the temperature did not alter the overall reaction path, it did impact the selectivity of each surface C<sub>1</sub> product by modifying the rate-determining steps and energy barriers associated with each C<sub>1</sub> product. Elevated temperatures might promote CO selectivity (with Ga<sub>2</sub>O<sub>3</sub> doping) and CH<sub>3</sub>OH/CH<sub>4</sub> selectivity (with La<sub>2</sub>O<sub>3</sub> doping).

# Temperature influence on $CO_2$ conversion with $H_2O$ at bifunctional sites via further loading transition metal clusters

During the photothermal catalytic reaction, the photoinduced oxygen vacancies on the surface of the oxide semiconductor were likely to be occupied in the intermediate steps of CO<sub>2</sub> conversion (especially in the step), which encompass the following steps: deoxygenation  $V_{O}-COOH \rightarrow COH'$ ,  $V_0^*CO_2 \rightarrow CO'$ ,  $V_{O}-HCOO \rightarrow CHO'$ ,  $V_{O}^{+}HCOOH \rightarrow HCOH',$  $V_{O}$ -<sup>\*</sup>HCOO→<sup>\*</sup>CHO',  $V_{O}-CO \rightarrow C'$ ,  $V_0$ -\*CHO $\rightarrow$ \*CH',  $V_0$ -\*HCHO $\rightarrow$ \*CH<sub>2</sub>', and  $V_0$ -\*CH<sub>3</sub>O $\rightarrow$ \*CH<sub>3</sub>'. Once the oxygen vacancies were consumed, the doped metal oxide was not conducive to the reduction reaction. As shown in Supplementary Fig. 7, it was calculated that at the M-V<sub>O</sub>-Ti (M=Al, La, Ga) site, oxygen vacancies were more prone to being occupied by intermediate species \*HCOOH, \*CO<sub>2</sub>, and \*CO<sub>2</sub> respectively to yield \*HCOH, \*CO, and \*CO. Subsequently, <sup>\*</sup>HCOH and <sup>\*</sup>CO underwent protonation. However, the energy barrier for the subsequent steps, specifically from <sup>\*</sup>CH<sub>2</sub>OH to form <sup>\*</sup>CH<sub>3</sub>OH or from <sup>\*</sup>CH<sub>3</sub> to form <sup>\*</sup>CH<sub>4</sub>, was exceedingly high. Consequently, the production of CH<sub>3</sub>OH and CH<sub>4</sub> becomes exceptionally challenging at this stage, underscoring the need for further enhancement of the reduction of key intermediates. This also showed that the introduction of adsorption-assisted sites primarily facilitates reactant adsorption, without necessarily guaranteeing the promotion of subsequent reduction reactions.

It was generally believed that transition metal clusters can activate  $H_2$  to  $*H^{52}$ , or promote the  $H_2O$  dissociation<sup>59</sup>, thereby providing \*H to facilitate

the direct hydrogenation of CO<sub>2</sub>. But at the same time, they might also catalyze the hydrogen evolution of H<sub>2</sub>O<sup>60,61</sup>, which engendered competition with hydrocarbon products during the reduction of CO<sub>2</sub> by H<sub>2</sub>O. Various transition metal clusters were further loaded onto Ga<sub>2</sub>O<sub>3</sub>-doped TiO<sub>2</sub> surface, to delve deeper into the conversion of CO2 with H2O into hydrocarbon products by introducing dual sites that assist in adsorption and reduction. In metal oxide semiconductor systems, the hydrogen evolution step was often identified as the rate-determining stage in the decomposition of H2O on the surface<sup>62,63</sup>. It was considered that a higher energy barrier for the hydrogen evolution step signifies effective suppression of competitive hydrogen evolution. The proton-donating process of water splitting was calculated on the surfaces supported by a series of transition metal clusters (Mn, Fe, Co, Ni, and Cu), including key steps such as water-splitting adsorption, H activation, and hydrogen evolution, as shown in Fig. 5. The results revealed that Mn, Fe, and Cu clusters, when loaded, exhibited lower energy barriers for the hydrogen evolution step compared to the pure aTiO<sub>2</sub> (101) surface, thereby promoting H<sub>2</sub> evolution. Furthermore, Mn and Fe impeded water splitting to yield <sup>\*</sup>H. In contrast, the inclusion of Ni- or Co-facilitated water splitting to produce <sup>\*</sup>H and activates H while discouraging the formation of H<sub>2</sub>. Therefore, we proceeded to calculate the path for CO<sub>2</sub> reduction to produce C1 on the surface of Ga2O3-doped TiO2 loaded with Ni or Co metal clusters. As shown in Fig. 6, it was found that, compared with the unloaded Ga<sub>2</sub>O<sub>3</sub>-doped TiO<sub>2</sub> surface, the energy barriers for the RDS in CH<sub>3</sub>OH and CH<sub>4</sub> production were significantly lowered when Ni or Co metal clusters were introduced. The impact of Ni was more significant than that of Co. In Comparison with the pure TiO<sub>2</sub> surface, Co could promote the formation of CH<sub>3</sub>OH and CH<sub>4</sub> primarily at higher temperatures. In addition, Co loading did not enhance the production of CO, and Ni loading only made the CO's RDS energy barrier lower when the temperature gradually increased (after 200 °C), with both Ni and Co improving the selectivity of CH<sub>3</sub>OH and CH<sub>4</sub>. This substantiated the positive role played by the supported Ni<sub>4</sub> or Co<sub>4</sub> cluster in facilitating the direct hydrogenation and multi-electron conversion of CO<sub>2</sub>. The electron transfer by Bader charge analysis has been investigated to further explore the possible mechanism and synergistic effect of Ga<sub>2</sub>O<sub>3</sub> doping combined with Co or Ni loading on the CO<sub>2</sub> conversion with the H<sub>2</sub>O reaction process. The results in Table 1 demonstrated that on



Fig. 6 | Comparison of RDS energy barriers for CO, CH<sub>3</sub>OH, and CH<sub>4</sub> production on the surfaces before and after featuring bifunctional sites under different temperatures (100~400 °C). Each graph shows the RDS energy barriers of surface

sites (Ti-O-Ti, Ti-O-Ga, Ti-O-Ga-Co<sub>4</sub>, and Ti-O-Ga-Ni<sub>4</sub>) at one specific reaction temperature: **a** 100 °C, **b** 150 °C, **c** 200 °C, **d** 250 °C, **e** 300 °C, **f** 350 °C, and **g** 400 °C.

Ti-O-Ga··Co<sub>4</sub> and Ti-O-Ga··Ni<sub>4</sub> sites, there was a decrease in positive charge on H and C atoms, while the negative charge on O atoms of  $CO_2$  and those connected to H atoms increased. This alteration suggests an augmented electron transfer between the surface and H or  $CO_2$ , leading to an accumulation of electrons on H and  $CO_2$  species. Consequently, the adsorption of  $CO_2$  and H was enhanced, as shown in Supplementary Table 2. Furthermore, the longer  $CO_2$  bond lengths signify a reduction in C=O bond energy and the weakening of the C=O bond. Based on the collective

Surface sites	Charge on H and O		Charge on C, O, and CO <sub>2</sub>				Bond length/ Å
	н	0	С	01	02	CO2	C=01; C=02
Ti-O-Ti	+0.6501	-1.1461	+2.1122	-1.0922	-1.0023	+0.0177	1.21; 1.33
Ti-O-Ga	+0.6034	-1.1201	+2.1964	-0.8620	-0.8542	+0.4802	1.27; 1.26
Ti-O-Ga··Co <sub>4</sub>	+0.6158	-1.1517	+2.1367	-1.1224	-1.1307	-0.1164	1.28; 1.29
Ti-O-Ga··Ni <sub>4</sub>	+0.6265	-1.1626	+2.0792	-1.1367	-1.0951	-0.1525	1.22; 1.36

Table 1 | The Bader charge analysis on Ti-O-Ti, Ti-O-Ga, Ti-O-Ga $\cdot\cdot$ Co<sub>4</sub>, and Ti-O-Ga $\cdot\cdot$ Ni<sub>4</sub> sites after the adsorption of CO<sub>2</sub>, H, and corresponding calculated bond lengths

computational findings, it can be inferred that  $Ga_2O_3$  doping combined with Co or Ni loading facilitates  $CO_2$  adsorption and the proton-donating process involved in H<sub>2</sub>O splitting by providing additional electrons, thereby facilitating the  $CO_2$  conversion reaction.

As the temperature rose from 100 °C to 400 °C, combined with Supplementary Fig. 8, it was observed that the reaction path on Ti-O-Ga-M<sub>T</sub>4 site for yielding C1 product remained constant, while the rate-determining steps underwent modifications. On the Co-loaded surface, the RDS for CO production consistently involved CO desorption, with the energy barrier decreasing as temperature increased, but was still higher compared to other surfaces. The RDS for CH<sub>3</sub>OH production transitioned from CH<sub>3</sub>OH desorption to  ${}^{*}HCHO \rightarrow {}^{*}CH_{2}OH$ , and the energy barrier decreased with increasing temperature. For CH4 production, the RDS shifted from  $^{*}CH_{2}OH \rightarrow ^{*}CH_{2}$  to  $^{*}HCHO \rightarrow ^{*}CH_{2}OH$ . Initially, the energy barrier decreased with rising temperature, but it subsequently began to increase at significantly high temperatures (350 °C). On the Ni-loaded surface, the RDS for CO production changed from '\*COOH $\rightarrow$ \*CO' to CO<sub>2</sub> adsorption, and the energy barrier diminished as temperature increased. In contrast, the RDS for CH<sub>3</sub>OH and CH<sub>4</sub> production shifted from CH<sub>3</sub>OH desorption to 'HCOO $\rightarrow$  HCOOH' and then to CO<sub>2</sub> adsorption, resulting in a larger energy barrier with increasing temperature. Irrespective of temperature, the RDS energy barrier for CH<sub>3</sub>OH and CH<sub>4</sub> production on the Ni-loaded surface was always lower than any other surface. The coexistence of dual sites on the surface resulted in CH<sub>3</sub>OH and CH<sub>4</sub> dominating the product selectivity even at very high temperatures. In contrast to surfaces solely doped with M2O3, wherein only adsorption-assisted sites were introduced, on these surfaces, as temperature increased, the energy barrier for the RDS in CO production remained persistently high, like the production of CH<sub>3</sub>OH and CH<sub>4</sub>. Alternatively, it gradually decreased until it governed C<sub>1</sub> product selectivity. The dual-site configuration, combining adsorption and reduction sites, not only accelerated the multi-electron conversion of CO<sub>2</sub> to generate CH<sub>3</sub>OH and CH<sub>4</sub> but also reduced temperature sensitivity, that is, CH<sub>3</sub>OH and CH<sub>4</sub> selectivity was still very high even at very high temperatures.

In this study, how temperature affects the complex intermediate process of photothermal catalytic CO2/H2O reaction and C1 product selectivity was investigated by employing periodic density functional theory methods. The surface reaction path and mechanism before and after regulating the active site were analyzed from the atomic-level perspective. It was revealed that on the pure TiO<sub>2</sub> semiconductor surface, the adsorption of reactants consistently emerged as the rate-determining step for C<sub>1</sub> production at high temperatures and that high temperatures did not favor the exothermic adsorption process. After adsorption activation, the subsequent conversion of CO<sub>2</sub> and H<sub>2</sub>O on the semiconductor surface involved multiple surface intermediates and multiple hydrodeoxygenation steps, resulting in the formation of a complex reaction network and potential energy surface. To establish a more universally applicable conclusion regarding the impact of temperature on the thermodynamics of surface reactions, we introduced diverse adsorption sites by incorporating a range of metal oxides onto the TiO<sub>2</sub> surface. Our investigations unveiled that during the photothermal conversion of CO2, Ti-O-M sites could always enhance CO2 adsorption, but did not invariably promote subsequent reduction reactions. Temperature variations did not alter the overall direction of the reaction path but brought

about alterations in the rate-determining step and the corresponding energy barrier for each  $C_1$  product, thereby changing the selectivity of each  $C_1$ product on the surface which followed the adjustment of active sites. Elevating the temperature might lead to an increased CO selectivity at the Ti-O-Ga site and a higher propensity for CH<sub>4</sub> and CH<sub>3</sub>OH selectivity at the Ti-O-La site. Upon introducing bifunctional sites (Ga-O-Ti-M<sub>T</sub>4) to facilitate adsorption and reduction, the production of CH<sub>3</sub>OH and CH<sub>4</sub> could be further enhanced through the facilitation of the proton donation process of H<sub>2</sub>O. However, it was noteworthy that as the temperature increases, it might become less favorable at the Ga-O-Ti-Ni4 site and might favor an increased selectivity for CO. While higher temperatures could directly augment hydrocarbon product generation at the Ga-O-Ti-Co4 site, there might be an inflection point where the rate-determining step's energy barrier started to rise. From a thermodynamic perspective, an increase in temperature did promote the reaction within a certain range, yet there might exist an optimal reaction temperature. The temperature elevation enhanced the reaction thermodynamics of C1 product formation to varying degrees at different surface sites.

Understanding the surface thermodynamic reaction mechanism affected by temperature and the roles it plays in the adsorption and desorption of reactants, intermediates, and products should be addressed. Here, the present first-principles computational work is complementary to experimentally elucidate the processes promoted by solar-induced temperature, and might provide valuable theoretical insights to guide active site regulation and further experimental design of photothermal catalysts dedicated to the conversion of  $CO_2$  and  $H_2O$  into hydrocarbons.

## Methods

#### **Computational methods**

All calculations have been conducted in the framework of DFT<sup>64,65</sup> using the Perdew–Burke–Ernzerhof (PBE) functional of the generalized gradient approximation (GGA)<sup>66</sup> as implemented in the Vienna Ab-initio Simulation Package (VASP)<sup>67,68</sup>. The core-valence electron interaction was described by the project-augmented wave (PAW) pseudopotential. The empirical correction in the DFT-D3 method with Becke–Jonson damping<sup>69</sup> was used for long-range dispersion corrections. To balance computational accuracy and cost, the energy cutoff of 400 eV was set in the calculations, along with the convergence threshold for the self-consistent-field iteration set to  $10^{-6}$  eV. Geometry optimizations converged when the force components were less than 0.01 eV/Å. The integration in the Brillouin zone was done on a  $3 \times 3 \times 1$  k-point grid sampled by the Gamma-centered scheme<sup>70</sup>.

The adsorption energy of CO<sub>2</sub>, H<sub>2</sub>O, and H, labeled as  $E_{ads}$ -CO<sub>2</sub>,  $E_{ads}$ -H<sub>2</sub>O, and  $E_{ads-H}$  were calculated as  $E_{ads} = E(\text{total}) - E(\text{slab}) - E(\text{adsorbed substance})$ . To study how temperature influences CO<sub>2</sub> conversion with H<sub>2</sub>O from a thermodynamic point of view, the Gibbs free energy ( $\Delta G$ ) of all intermediates during the reaction were calculated based on statistical thermodynamics<sup>71–73</sup> by adding the importance of entropic contribution (S) and zero-point energy (ZPE) corrections as:

$$\Delta G = \Delta H - T \Delta S \tag{1}$$

$$\Delta H \approx \Delta U = \Delta E + \Delta ZPE + \Delta U_{tran} + \Delta U_{rot} + \Delta U_{vib}$$
(2)

$$\Delta S = \Delta S_{tran} + \Delta S_{rot} + \Delta S_{vib}$$
(3)

Where  $\Delta E$  is the reaction energy acquired from DFT calculations of the optimized structure of reactants and products at each intermediate step, calculated as  $\Delta E = E(\text{products}) - E(\text{reactants})$ .  $\Delta ZPE$  is the difference in zero-point energy due to the reaction.  $\Delta U$  and  $\Delta S$  are the change in internal energy and entropy, contributed by the translation, rotation, and vibration, respectively, which could be given by:

$$U_{\text{tran}} = U_{\text{rot}} = \frac{3}{2}k_B T \tag{4}$$

$$U_{\rm vib} = \sum_{\rm i} \frac{h\nu_i}{\left(e^{h\nu_i/k_B T} - 1\right)} \tag{5}$$

$$S_{\text{tran}} = k_B \left\{ \ln \left[ \left( \frac{2\pi m k_B T}{h^2} \right)^{3/2} \frac{k_B T}{P} \right] + \frac{5}{2} \right\}$$
(6)

$$S_{\rm rot} = k_B \left\{ \ln \left[ \frac{\sqrt{\pi I_A I_B I_C}}{\sigma} \left( \frac{8\pi^2 k_B T}{h^2} \right)^{3/2} \right] + \frac{3}{2} \right\}$$
(7)

$$S_{\text{vib}} = k_B \sum_{i} \left( \frac{hv_i}{k_B T \left( e^{hv_i/k_B T} - 1 \right)} - \ln \left( 1 - e^{-hv_i/k_B T} \right) \right)$$
(8)

Where *T* is temperature,  $v_i$  are vibrational frequencies that could be calculated by DFT calculations. Then  $\Delta ZPE$ ,  $\Delta U$ , and  $\Delta S$  are determined from  $v_i$  associated with the normal modes of the adsorbed species calculated in the harmonic oscillator approximation, which can be calculated using VASPKIT<sup>74</sup> after vibration frequency calculation. The temperature (*T* = 298.15, 373.15, 423.15, 473.15, 523.15, 573.15, 623.15, 673.15 K) dependence of entropic and vibrational contributions to the Gibbs free energy has been explicitly taken into account. It is worth mentioning that this computational strategy for analyzing the influence of temperature effects on surface reaction mechanisms from the thermodynamics perspective has been applied to some reported research in recent years<sup>62,75,76</sup>.

#### **Computational models**

The modified metal oxide semiconductor systems were constructed based on our previous theoretical and experimental studies<sup>56,77,78</sup>. Typically, TiO<sub>2</sub>based catalysts can maintain a stable crystal form of anatase at temperatures below approximately 450 °C. Furthermore, it has been observed that a temperature increase within this range results in a slight decrease in the band gap of TiO<sub>2</sub> due to the expansion of atomic spacing, though this influence is almost negligible<sup>23</sup>. In the meantime, taking the temperature range employed in our photothermal catalysis experiments into account, the temperature range of this study is set as 298.15~673.15 K. Consequently, the supporting catalyst, that is, the pure anatase TiO2, whose most stable surface - (101) surface was directly introduced into the calculation. The  $aTiO_2(101)$ surface slab was optimized with three layers in a 4 × 4 supercell (144 atoms), keeping fixed the bottom layer to the bulk atomic positions and the top two layers allowed to relax. A vacuum space of 15 Å along the surface normal direction is applied to minimize unphysical interaction between periodic images and simulate the exposed surface. One bridge O atom was removed to represent the defective surface with oxygen vacancy (V<sub>O</sub>). Furthermore, according to the reported literature, for the metal oxides (M<sub>x</sub>O<sub>y</sub>) decorating surface, the model of M<sub>x</sub>O<sub>y</sub> doped TiO<sub>2</sub> catalysts was built by incorporating one metal atom to replace one five-coordination Ti atom<sup>79-81</sup>. And as the previously extensively accepted 4-atom cluster model catalyst  $^{73,82-84},$  the  $M_{\rm T}4$ cluster consists of four M<sub>T</sub> atoms that were put onto the supporting catalyst surface, to signify the transition metal (M<sub>T</sub>) loaded surface. The most stable adsorption configurations of all reaction intermediates at the surface reaction sites, were identified by the lowest calculated energy values among all possible configurations. The catalysts' models and the optimized reaction intermediate structures on corresponding surface-active sites can be seen in Supplementary Figs. 1 and 2, respectively.

#### **Data availability**

The authors declare that the data supporting the findings of this study are available within the article and its supplementary information files or from the corresponding authors on reasonable request.

#### Code availability

The VASP calculation inputs and outputs will be shared with the community upon reasonable request.

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# **Author contributions**

L.Z.: Conceptualization; methodology; validation; formal analysis; investigation; writing original draft; visualization. C.-Q. L.: Methodology; investigation. Y.L.: Formal analysis; investigation. C.-Y. X.: Revision; feedback; editing. Y.-W. Z.: Resources; project administration; funding acquisition. All authors have given approval to the final version of the manuscript.

## **Competing interests**

The authors declare no competing interests.

# Additional information

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